

## The Crystal Structure of $\text{Os}_4\text{Al}_{13}$

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The crystal structure of  $\text{Os}_4\text{Al}_{13}$  has been determined and refined from three-dimensional X-ray data. The monoclinic cell dimensions are  $a = 17.64$ ,  $b = 4.228$ ,  $c = 7.773$  Å and  $\beta = 115.15^\circ$ . The space-group is  $C2/m$ . The atomic arrangement is related to that present in  $\text{Fe}_4\text{Al}_{13}$ . The interatomic distances in the two structures are compared.

Very little is known about the platinum metal aluminides rich in aluminum. An investigation of such alloys has been started at this Institute and has so far revealed the existence of several new phases, complex in structure and with similarities to earlier known structures like  $\text{Fe}_4\text{Al}_{13}$ ,<sup>1,2</sup>  $\text{Co}_4\text{Al}_{13}$ ,<sup>3</sup> and  $\text{Ni}_4\text{Mn}_{11}\text{Al}_{60}$ .<sup>4</sup> The simplest structure so far known in this group is represented by  $\text{Os}_4\text{Al}_{13}$ .

### EXPERIMENTAL

Osmium-aluminum alloys were prepared by melting pellets, obtained by pressing osmium powder (L. Light and Co., 99.975 %) and pieces of aluminum (E. Merck A.G., at least 99.99 % pure), in an electric arc furnace under an argon pressure of 0.5 atm. The loss of aluminum from vaporization during the short melting period was determined by weighing the sample before and after the heat treatment. By comparing the X-ray powder patterns of samples of different compositions, the phase was found to exist in a pure state with a formula slightly higher in aluminum than  $\text{OsAl}_3$ . This preparation is grey and crystalline and contains well formed prismatic crystals very often forming aggregates. After having tried several crystals found to be twinned, a single crystal was eventually picked up from the crushed melt. The form of this crystal, which was used for the structure determination, was nearly a rectangular parallelepiped with the edges approximately 0.004, 0.008, and 0.02 mm in length. Weissenberg photographs were taken with rotation around an axis parallel to the longest edge of the crystal.  $\text{CuK}$  radiation was used and three layers were registered by the multiple-film technique. A visual estimate of 380 independent intensities was made by comparison with an intensity scale made from a suitable reflection on the zero layer.

The computational work involved in this study was to a large extent carried out using the computers BESK and FACIT EDB with the help of several computer programs.<sup>5</sup> The procedure of refinement by the least-squares method has been briefly described elsewhere.<sup>6</sup>

## DERIVATION OF THE STRUCTURE

The Weissenberg photographs showed monoclinic symmetry. Guinier powder photographs, registered with monochromatized  $\text{CuK}\alpha_1$  radiation and using KCl ( $a = 6.2919 \text{ \AA}$  at  $20^\circ\text{C}$ ) as an internal standard, gave the following monoclinic cell dimensions in  $\text{\AA}$ :

$$\begin{aligned} a_m &= 17.64 \pm 0.01 \\ b_m &= 4.228 \pm 0.002 & \beta_m &= 115.15 \pm 0.04^\circ \\ c_m &= 7.773 \pm 0.002 \end{aligned}$$

Weissenberg photographs of the twinned crystals often looked orthorhombic. The unit cell may be described using an alternative coordinate system which reflects the pseudo-orthorhombic character actually existing in the structure:

$$\begin{aligned} a_o &= 15.88 \pm 0.02 \\ b_o &= 4.228 \pm 0.002 & \beta_o &= 91.4 \pm 0.1^\circ \\ c_o &= 7.773 \pm 0.002 \end{aligned}$$

The relation between the two representations of the unit are:

$$\begin{aligned} \vec{a}_o &= \vec{c}_m + \vec{a}_m \\ \vec{b}_o &= \vec{b}_m \\ \vec{c}_o &= -\vec{c}_m \end{aligned}$$

The reflections observed are in agreement with a highest symmetry of  $C2/m$ . The positions of the heavy atoms were found in the Patterson projection along  $[010]$ . These positions corresponding to an  $R$ -value of 0.26 between

Table 1. Weight analysis in Os<sub>4</sub>Al<sub>13</sub>

$\sin \Theta$ -interval	$\overline{w \Delta^2}$	number of reflections in each interval
0.00—0.46	1.19	73
0.46—0.58	1.03	52
0.58—0.66	1.25	51
0.66—0.73	0.80	43
0.73—0.79	1.13	31
0.79—0.84	0.97	41
0.84—0.88	0.88	32
0.88—0.92	0.55	27
0.92—0.96	0.35	14
0.96—0.99	1.16	16
$F_o$ -interval		
0 — 58	0.77	91
58 — 115	1.50	80
115 — 173	0.89	79
173 — 230	0.66	46
230 — 288	1.18	45
288 — 345	1.06	19
345 — 505	0.80	20

Table 2. Final atomic parameters and their e.s.d.'s

Atom	$x$	$\sigma(x)$ Å	$y$	$z$	$\sigma(z)$ Å	$B$ Å <sup>2</sup>	$\sigma(B)$ Å <sup>2</sup>
Os <sub>1</sub>	0.2940	0.0001	0	0.2915	0.0003	0.22	0.03
Os <sub>2</sub>	0.4919	0.0001	$\frac{1}{2}$	0.1947	0.0003	0.22	0.03
Al <sub>1</sub>	0.087	0.001	$\frac{1}{2}$	0.368	0.002	1.42	0.28
Al <sub>2</sub>	0.257	0.001	0	0.613	0.002	1.00	0.26
Al <sub>3</sub>	0.132	0.001	0	0.162	0.002	0.99	0.26
Al <sub>4</sub>	0.290	0.001	$\frac{1}{2}$	0.087	0.002	1.60	0.29
Al <sub>5</sub>	0.414	0.001	$\frac{1}{2}$	0.432	0.002	0.95	0.25
Al <sub>6</sub>	0.409	0.001	0	0.194	0.002	1.60	0.28
Al <sub>7</sub>	0		$\frac{1}{2}$	0		0.68	0.33

the  $F_o(h0l)$  and  $F_c(h0l)$  values, gave a starting point for the derivation of the electron density projection  $\rho(xz)$ . In this way it was eventually possible to locate the sites of 8 osmium and 26 aluminium atoms all situated on mirror planes and to determine approximate positional parameters for these atoms. The suggested structure, containing two formula units  $\text{Os}_4\text{Al}_{13}$ , was then refined by least-squares method.

The weights were calculated using Hughes' weighting scheme. The final  $R$ -value is 0.103. The weight analysis obtained in the last cycle of refinement

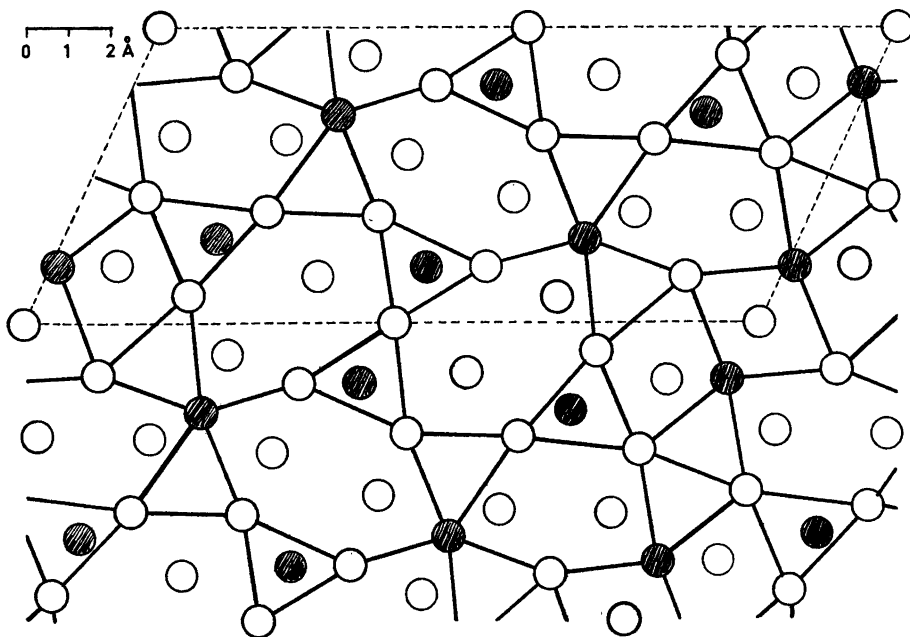


Fig. 1. The crystal structure of  $\text{Os}_4\text{Al}_{13}$ , seen along the  $b$ -axis. Connected atoms at  $y = 0$  and unconnected at  $y = \frac{1}{2}$ . Os-atoms are shaded.

Table 3. Observed and calculated structure factors.

$k = 0$						
$h$	$l$	$F_o$	$F_c$	$h$	$l$	$F_c$
2	0	37	34	$\overline{10}$	3	81
4	0	298	335	$\overline{8}$	3	-322
6	0	131	126	$\overline{6}$	3	12
8	0	95	106	$\overline{4}$	3	-139
10	0	405	408	$\overline{2}$	3	-248
12	0	< 29	27	0	3	-49
14	0	289	274	2	3	-398
16	0	30	45	4	3	49
18	0	< 30	14	6	3	-275
20	0	145	155	8	3	-153
$\overline{20}$	1	182	-190	10	3	-97
$\overline{18}$	1	136	130	12	3	-227
$\overline{16}$	1	195	-189	14	3	-13
$\overline{14}$	1	43	37	16	3	-192
$\overline{12}$	1	114	-128	$\overline{20}$	4	70
$\overline{10}$	1	87	-94	$\overline{18}$	4	252
$\overline{8}$	1	259	247	$\overline{16}$	4	9
$\overline{6}$	1	185	-165	$\overline{14}$	4	360
$\overline{4}$	1	241	235	$\overline{12}$	4	43
$\overline{2}$	1	15	-14	$\overline{10}$	4	90
0	1	18	17	$\overline{8}$	4	136
2	1	226	292	$\overline{6}$	4	-17
4	1	83	-86	$\overline{4}$	4	384
6	1	434	460	2	4	-55
8	1	< 24	10	0	4	62
10	1	144	157	2	4	44
12	1	165	170	4	4	-157
14	1	44	53	6	4	194
16	1	279	278	8	4	-169
$\overline{20}$	2	144	-146	10	4	127
$\overline{18}$	2	139	-142	12	4	-121
$\overline{16}$	2	< 31	-26	14	4	-111
$\overline{14}$	2	356	-360	$\overline{22}$	5	234
$\overline{12}$	2	< 27	5	$\overline{20}$	5	16
$\overline{10}$	2	294	-295	$\overline{18}$	5	135
$\overline{8}$	2	126	-112	$\overline{16}$	5	103
$\overline{6}$	2	175	-171	$\overline{14}$	5	19
$\overline{4}$	2	359	-364	$\overline{12}$	5	410
$\overline{2}$	2	35	24	$\overline{10}$	5	3
0	2	362	-390	$\overline{8}$	5	415
2	2	49	-38	$\overline{6}$	5	79
4	2	76	-79	$\overline{4}$	5	117
6	2	153	-162	$\overline{2}$	5	281
8	2	< 27	24	0	5	79
10	2	255	-222	2	5	420
12	2	104	98	4	5	46
14	2	< 29	-24	6	5	113
16	2	54	-39	8	5	171
$\overline{22}$	3	96	-81	10	5	6
$\overline{20}$	3	139	122	12	5	238
$\overline{18}$	3	109	-104	$\overline{18}$	6	-170
$\overline{16}$	3	< 31	21	$\overline{16}$	6	128
$\overline{14}$	3	< 29	-33	$\overline{14}$	6	-100
$\overline{12}$	3	156	-143	$\overline{12}$	6	2

$h$	$F_o$	$F_c$	$h$	$l$	$F_o$	$F_c$
$\overline{10}$ 6	60	50	$\overline{5}$ 1		126	87
$\overline{8}$ 6	119	-106	$\overline{3}$ 1		146	-171
$\overline{6}$ 6	330	257	$\overline{1}$ 1		233	235
$\overline{4}$ 6	129	-125	1 1		111	-109
$\overline{2}$ 6	194	207	3 1		117	116
0 6	82	81	5 1		235	222
2 6	< 31	30	7 1		< 26	12
4 6	202	196	9 1		344	296
6 6	< 28	- 23	11 1		104	-106
8 6	258	240	13 1		301	255
$\overline{18}$ 7	114	-125	15 1		123	92
$\overline{16}$ 7	144	-166	17 1		84	86
$\overline{14}$ 7	43	- 48	$\overline{21}$ 2		264	-244
$\overline{12}$ 7	283	-304	$\overline{19}$ 2		29	- 34
$\overline{10}$ 7	< 30	0	$\overline{17}$ 2		329	-272
$\overline{8}$ 7	154	-149	$\overline{15}$ 2		104	-103
$\overline{6}$ 7	133	-135	$\overline{13}$ 2		102	- 92
$\overline{4}$ 7	54	- 60	$\overline{11}$ 2		304	-315
$\overline{2}$ 7	217	-236	$\overline{9}$ 2		< 25	6
0 7	30	46	$\overline{7}$ 2		505	-603
2 7	153	-151	$\overline{5}$ 2		72	- 49
$\overline{18}$ 8	89	97	$\overline{3}$ 2		196	-201
$\overline{16}$ 8	90	-102	$\overline{1}$ 2		172	-149
$\overline{14}$ 8	55	- 62	1 2		65	- 58
$\overline{12}$ 8	< 29	- 30	3 2		376	-463
$\overline{10}$ 8	195	-187	5 2		< 25	10
$\overline{8}$ 8	79	57	7 2		235	-196
$\overline{6}$ 8	232	-229	9 2		56	- 48
$\overline{4}$ 8	< 29	- 25	11 2		78	127
$\overline{2}$ 8	107	-114	13 2		268	-189
0 8	189	-185	15 2		112	89
$\overline{16}$ 9	171	198	17 2		176	-161
$\overline{14}$ 9	< 22	15	$\overline{15}$ 3		328	-273
$\overline{12}$ 9	196	193	$\overline{13}$ 3		< 32	42
$\overline{10}$ 9	< 24	33	$\overline{11}$ 3		222	-180
$\overline{8}$ 9	< 34	- 29	$\overline{9}$ 3		59	- 59
$\overline{6}$ 9	126	132	$\overline{7}$ 3		< 23	- 15
$\overline{4}$ 9	75	- 70	$\overline{5}$ 3		356	-402
$\overline{2}$ 9	188	185	$\overline{3}$ 3		70	46
			$\overline{1}$ 3		434	-515
	$k = 1$		1 3		40	- 41
1 0	155	168	3 3		143	-156
3 0	291	360	5 3		254	-230
5 0	< 18	18	7 3		79	- 80
7 0	376	411	9 3		433	-380
9 0	27	49	11 3		< 34	- 32
11 0	135	114	13 3		173	-172
13 0	160	163	15 3		155	-138
15 0	< 34	25	$\overline{21}$ 4		265	248
17 0	231	202	$\overline{19}$ 4		< 30	- 18
$\overline{19}$ 1	130	-180	$\overline{17}$ 4		190	200
$\overline{17}$ 1	113	-108	$\overline{15}$ 4		69	78
$\overline{15}$ 1	< 34	48	$\overline{13}$ 4		74	82
$\overline{13}$ 1	287	-257	$\overline{11}$ 4		296	258
$\overline{11}$ 1	102	95	$\overline{9}$ 4		109	- 94
$\overline{9}$ 1	125	-106	$\overline{7}$ 4		267	221
$\overline{7}$ 1	< 21	2	$\overline{5}$ 4		< 25	6
			$\overline{3}$ 4		< 25	19

$h$	$l$	$F_o$	$F_c$	$h$	$l$	$F_o$	$F_c$
						$k = 2$	
$\bar{1}$	4	145	137	2	0	< 38	28
1	4	147	-141	4	0	244	275
3	4	206	190	6	0	105	113
5	4	198	-170	8	0	92	91
7	4	< 34	- 33	10	0	335	357
9	4	< 34	4	12	0	37	21
11	4	136	-129	14	0	265	248
$\bar{19}$	5	164	168	16	0	32	42
$\bar{17}$	5	< 33	- 33	$\bar{18}$	1	136	115
$\bar{15}$	5	240	272	$\bar{16}$	1	198	-172
$\bar{13}$	5	< 34	40	$\bar{14}$	1	< 26	32
$\bar{11}$	5	153	140	$\bar{12}$	1	108	-109
$\bar{9}$	5	208	193	$\bar{10}$	1	86	- 85
$\bar{7}$	5	90	75	$\bar{8}$	1	233	210
$\bar{5}$	5	363	373	$\bar{6}$	1	134	-140
$\bar{3}$	5	< 30	2	$\bar{4}$	1	175	189
$\bar{1}$	5	201	219	$\bar{2}$	1	< 10	- 9
1	5	160	158	0	1	< 10	13
3	5	90	91	2	1	208	237
5	5	245	242	4	1	73	- 73
7	5	< 34	16	6	1	320	391
9	5	198	186	8	1	< 23	8
$\bar{21}$	6	164	-174	10	1	133	141
$\bar{19}$	6	< 27	- 15	12	1	167	158
$\bar{17}$	6	< 31	- 29	14	1	< 74	45
$\bar{15}$	6	68	- 90	16	1	266	254
$\bar{13}$	6	69	83	$\bar{20}$	2	137	-136
$\bar{11}$	6	217	-210	$\bar{18}$	2	138	-132
$\bar{9}$	6	117	86	$\bar{16}$	2	< 25	- 24
$\bar{7}$	6	< 33	34	$\bar{14}$	2	334	-326
$\bar{5}$	6	< 33	14	$\bar{12}$	2	< 25	- 1
$\bar{3}$	6	199	183	$\bar{10}$	2	232	-260
$\bar{1}$	6	145	-129	$\bar{8}$	2	109	-101
1	6	217	220	$\bar{6}$	2	146	-141
3	6	< 34	- 43	$\bar{4}$	2	285	-303
5	6	140	127	$\bar{2}$	2	20	24
7	6	175	164	0	2	261	-321
$\bar{19}$	7	304	-302	2	2	34	- 33
$\bar{17}$	7	< 29	- 19	4	2	66	- 70
$\bar{15}$	7	243	-247	6	2	160	-145
$\bar{13}$	7	101	- 89	8	2	< 25	27
$\bar{11}$	7	91	- 70	10	2	215	-202
$\bar{9}$	7	247	-259	12	2	88	89
$\bar{7}$	7	34	- 42	14	2	< 22	- 25
$\bar{5}$	7	268	-322	16	2	39	- 37
$\bar{3}$	7	34	- 48	$\bar{20}$	3	123	114
$\bar{1}$	7	< 33	2	$\bar{18}$	3	96	- 97
1	7	171	-168	$\bar{16}$	3	< 25	20
$\bar{17}$	8	93	- 99	$\bar{14}$	3	< 26	- 32
$\bar{15}$	8	56	61	$\bar{12}$	3	134	-130
$\bar{13}$	8	240	-269	$\bar{10}$	3	84	73
$\bar{11}$	8	31	42	$\bar{8}$	3	279	-279
$\bar{9}$	8	161	-167	$\bar{6}$	3	< 19	11
$\bar{7}$	8	85	-108	$\bar{4}$	3	136	-123
$\bar{5}$	8	31	- 42	$\bar{2}$	3	181	-209
$\bar{3}$	8	233	-238				
$\bar{1}$	8	180	183				

$h$	$l$	$F_o$	$F_c$	$h$	$l$	$F_o$	$F_c$
0	3	38	-41	8	5	169	156
2	3	279	-344	$\overline{18}$	6	164	-156
4	3	32	42	$\overline{16}$	6	119	115
6	3	234	-248	$\overline{14}$	6	98	-92
8	3	158	-139	$\overline{12}$	6	< 26	4
10	3	89	-88	$\overline{10}$	6	65	47
12	3	227	-208	$\overline{8}$	6	109	-97
$\overline{20}$	4	59	67	$\overline{6}$	6	220	231
$\overline{18}$	4	212	228	$\overline{4}$	6	129	-111
$\overline{16}$	4	< 26	9	$\overline{2}$	6	179	185
$\overline{14}$	4	289	324	0	6	69	74
$\overline{12}$	4	< 25	37	2	6	< 25	25
$\overline{10}$	4	100	84	4	6	196	181
$\overline{8}$	4	165	122	6	6	< 20	-21
$\overline{6}$	4	< 22	-17	8	6	263	221
$\overline{4}$	4	332	329	$\overline{18}$	7	102	-116
$\overline{2}$	4	53	-41	$\overline{16}$	7	140	-154
0	4	72	58	$\overline{14}$	7	32	-40
2	4	54	39	$\overline{12}$	7	270	-275
4	4	150	-137	$\overline{10}$	7	< 26	0
6	4	183	172	$\overline{8}$	7	138	-139
8	4	191	-155	$\overline{6}$	7	134	-125
10	4	128	112	$\overline{4}$	7	57	-51
12	4	113	-110	$\overline{2}$	7	195	-215
$\overline{18}$	5	128	126	0	7	23	42
$\overline{16}$	5	89	96	2	7	147	-139
$\overline{14}$	5	< 26	20	$\overline{16}$	8	87	-97
$\overline{12}$	5	333	366	$\overline{14}$	8	54	-55
$\overline{10}$	5	26	2	$\overline{12}$	8	< 21	-27
$\overline{8}$	5	360	366	$\overline{10}$	8	178	-171
$\overline{6}$	5	92	73	$\overline{8}$	8	54	52
$\overline{4}$	5	127	107	$\overline{6}$	8	196	-210
$\overline{2}$	5	264	252	$\overline{4}$	8	< 21	-24
0	5	81	67	$\overline{2}$	8	97	-106
2	5	341	376	0	8	179	-170
4	5	52	43	$\overline{12}$	9	136	179
6	5	109	106				

is given in Table 1 and the final atomic parameters and their e.s.d.'s are given in Table 2. The observed and calculated  $F$ -values are given in Table 3.

#### THE STRUCTURE

The structure thus derived is built up of plane nets of atoms, illustrated by the connected atoms at  $y = 0$  in Fig. 1. The nets are repeated along [010] by a  $2_1$  axis of  $C2/m$ .

The interatomic distances shorter than 3.5 Å are listed in Table 4. The mean distances and the coordinations around the various atoms are shown in Table 5, atoms at larger distances than 3 Å being disregarded.

The standard deviations in the interatomic distances are:

Os—Al <sub>7</sub>	0.004 Å
Os—Al <sub>1-6</sub>	0.03 Å
Al <sub>7</sub> —Al <sub>1-6</sub>	0.03 Å
Al <sub>1-6</sub> —Al <sub>1-6</sub>	0.04 Å

Table 4. Interatomic distances in Os<sub>4</sub>Al<sub>13</sub> (Å).

Os <sub>1</sub> — Al <sub>6</sub>	2.457	Os <sub>2</sub>	2.587
2Al <sub>2</sub>	2.531	Os <sub>2</sub>	2.675
Al <sub>1</sub>	2.565	Al <sub>4</sub>	2.825
Al <sub>3</sub>	2.586	2Al <sub>2</sub>	2.894
2Al <sub>4</sub>	2.627	2Al <sub>1</sub>	2.962
Al <sub>4</sub>	2.654	2Al <sub>7</sub>	2.983
Al <sub>2</sub>	2.846	(2Al <sub>6</sub> )	3.299
2Al <sub>5</sub>	2.855	Al <sub>4</sub> — 2Os <sub>1</sub>	2.627
Os <sub>2</sub> — 2Al <sub>6</sub>	2.570	Os <sub>1</sub>	2.654
Al <sub>3</sub>	2.587	2Al <sub>4</sub>	2.573
Al <sub>5</sub>	2.637	Al <sub>5</sub>	2.623
2Al <sub>7</sub>	2.639	Al <sub>2</sub>	2.794
2Al <sub>1</sub>	2.667	Al <sub>3</sub>	2.825
Al <sub>3</sub>	2.675	2Al <sub>6</sub>	2.835
Al <sub>5</sub>	2.738	(Al <sub>6</sub> )	3.221
(Os <sub>2</sub> )	3.161	(Os <sub>2</sub> )	3.285
(Al <sub>4</sub> )	3.285	Al <sub>5</sub> — Os <sub>2</sub>	2.637
Al <sub>1</sub> — Os <sub>1</sub>	2.565	Os <sub>2</sub>	2.738
2Os <sub>2</sub>	2.667	2Os <sub>1</sub>	2.855
Al <sub>7</sub>	2.588	Al <sub>4</sub>	2.623
2Al <sub>5</sub>	2.628	2Al <sub>1</sub>	2.628
Al <sub>2</sub>	2.693	Al <sub>5</sub>	2.733
Al <sub>6</sub>	2.823	2Al <sub>6</sub>	2.787
2Al <sub>3</sub>	2.962	Al <sub>2</sub>	2.888
(Al <sub>6</sub> )	3.376	Al <sub>6</sub> — Os <sub>1</sub>	2.457
(2Al <sub>4</sub> )	3.485	2Os <sub>2</sub>	2.570
Al <sub>2</sub> — 2Os <sub>1</sub>	2.531	Al <sub>7</sub>	2.639
Os <sub>1</sub>	2.846	2Al <sub>5</sub>	2.787
2Al <sub>2</sub>	2.693	Al <sub>1</sub>	2.823
Al <sub>1</sub>	2.693	2Al <sub>2</sub>	2.835
Al <sub>4</sub>	2.794	(Al <sub>4</sub> )	3.221
Al <sub>5</sub>	2.888	(Al <sub>1</sub> )	3.376
2Al <sub>3</sub>	2.894	Al <sub>7</sub> — 4Os <sub>2</sub>	2.639
(Al <sub>3</sub> )	3.232	2Al <sub>1</sub>	2.588
(2Al <sub>1</sub> )	3.485	2Al <sub>6</sub>	2.639
Al <sub>3</sub> — Os <sub>1</sub>	2.586	2Al <sub>3</sub>	2.983

Table 5. Coordination and mean values of interatomic distances (in Å) in Os<sub>4</sub>Al<sub>13</sub>.

	Os <sub>1</sub> — 11 Al	2.649		
	Os <sub>2</sub> — 10 Al	2.639		
Al <sub>1</sub> — 3 Os	2.633	— 7 Al	2.755	
Al <sub>2</sub> — 3 Os	2.636	— 7 Al	2.793	
Al <sub>3</sub> — 3 Os	2.616	— 7 Al	2.929	
Al <sub>4</sub> — 3 Os	2.646	— 7 Al	2.723	
Al <sub>5</sub> — 4 Os	2.771	— 7 Al	2.725	
Al <sub>6</sub> — 3 Os	2.532	— 6 Al	2.784	
Al <sub>7</sub> — 4 Os	2.639	— 8 Al	2.798	
Al— Os mean dist.	2.645	Al—Al mean dist.	2.787	

The Os—Al distances are in the range 2.46—2.86 Å and the Al—Al distances in the range 2.57—2.96 Å. The mean values of the Os—Al and Al—Al distances are 2.65 Å and 2.78 Å. These mean distances are in excellent concordance with those found in Fe<sub>4</sub>Al<sub>13</sub>, viz. 2.55 Å for the Fe—Al distances and 2.77



Å for the Al—Al distances, taking into consideration the fact that the radius of Os should be about 0.1 Å larger than that of Fe. It is, however, worth mentioning that no extremely short distances, as those found for some Fe—Al and Al—Al contacts in  $\text{Fe}_4\text{Al}_{13}$ , are present in  $\text{Os}_4\text{Al}_{13}$ . Thus while in the former compound the Al—Al distances fall into two ranges (2.60—2.75 Å and 2.83—2.95 Å), a similar division of these distances is not present in the osmium phase.

This structure will be further discussed when other structures of this type, now under study, have been determined.

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